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Original Citation:

Two-Photon Lithography of 3D Nanocomposite Piezoelectric Scaffolds for Cell Stimulation / Marino Attilio; Barsotti Jonathan; de Vito G; Filippeschi Carlo; Mazzolai Barbara; Piazza Vincenzo; Labardi Massimiliano; Mattoli Virgilio; Ciofani Gianni. - In: ACS APPLIED MATERIALS & INTERFACES. - ISSN 1944-8252. - ELETTRONICO. - 7:(2015), pp. 25574-25579. [10.1021/acsami.5b08764]

Availability:

This version is available at: 2158/1175448 since: 2020-05-02T17:27:08Z

Published version:

DOI: 10.1021/acsami.5b08764

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Two-Photon Lithography of 3D Nanocomposite Piezoelectric Scaffolds for Cell Stimulation

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ABSTRACT

In this letter we report on the fabrication, the characterization, and the *in vitro* testing of structures suitable for cell culturing, prepared through two-photon polymerization of a nanocomposite resist. More in details, commercially-availableOrmocomp® has been doped with piezoelectric barium titanate nanoparticles, and bioinspired 3D structures resembling trabeculae of sponge bone have been fabricated. After an extensive characterization, preliminary *in vitro* testing demonstrated that both the topographical and the piezoelectric cues of these scaffolds are able to enhance the differentiation process of human SaOS-2 cells in terms of hydroxyapatite deposition.

KEYWORDS

Two-photon lithography; barium titanate nanoparticles; direct laser writing; piezoelectric stimulation; bone tissue engineering.

Two-photon lithography (TPL) is a disrupting technology that allows the fabrication of complex 3D nanostructured scaffolds, owing to the mechanism of two-photon absorption and polymerization of dedicated photoresists. ¹ Obtained 3D structures are suitable for *in vitro* testing on living cells, and can be exploited for the investigation of cell / substrate biophysical interactions, for the promotion of a specific cell phenotype, and for the modification of surfaces of biomedical devices. ^{2,3} As an example, concerning bone tissue engineering we have recently developed 3D structures bioinspired by the geometry of a sponge-bone trabecula, named “Osteo-

Prints”: these scaffolds have been demonstrated to promote the osteogenic differentiation of

SaOS-2 osteoblast-like cells through topographical stimulation.⁴ More in details, the enhanced osteogenesis was proven to be induced by the presence of the 3D biomimetic niches able to affect the cellular (and nuclear) shape and fostering cell commitment toward osteogenesis.

The availability of several materials that can be exploited as photo-resists for the TPL allows the control of a wide range of physical / chemical properties of the material (*i.e.*, stiffness, porosity, roughness, biodegradability, *etc.*), which can be further tuned by doping the resists with

appropriate nanomaterials (*i.e.*, single-walled carbon nanotubes,⁵ titanium dioxide nanoparticles,⁶

magnetic nanoparticles,⁷ piezoelectric nanoparticles,⁸ zinc oxide nanowires,⁹ *etc.*), thus obtaining

"smart" features not achievable by the use of the corresponding bulk material.¹⁰ It is well known that composite / nano-composite materials are gaining considerable interest in a wide range of research areas, from the aerospace field to bio-applications, because of the possibility of finely

tune their chemical and physical properties.¹¹ Further advancements in the biomedical field are expected, also owing to innovative fabrication approaches, that allow for rapid prototype

production.¹² Scaffolds prepared with such nano-composites can show active / sensitive behaviors, deriving, for example, by the magnetic, conductive, or piezoelectric properties of the exploited nanoparticles; obtained structures can thus be successfully exploited for an active stimulation of cells and, eventually, tissues.

As an example of peculiar 3D nano-composite structures fabricated by TPL, the Kawata group recently developed a technique to obtain nanocomposites in which the embedded single-walled

carbon nanotubes (SWCNTs) are aligned along a desired directions, *i.e.*, along the laser scanning guidance.⁵ Anisotropic structures characterized by aligned SWCNTs are extremely interesting in the field of actuators, besides exhibiting an enhancement of mechanical, electrical, thermal, and optical properties with respect to analogous isotropic constructs. Concerning magnetic structures fabricated by TPL, Wang *et al.* reported on the remote manipulation through an external ferromagnetic field of microspring-like shaped nanocomposites containing Fe₃O₄ nanoparticles.⁷ Further improvements in this approach allowed the magnetic control of 3D scaffolds fabricated by TLP suitable the stem cell transplanation applications,¹³ and of magnetic helical microswimmers for targeted gene delivery.¹⁴

Focusing on piezoelectric nanomaterials, different cell types (including neurons,¹⁵ neural stem cells,¹⁶ and fibroblasts¹⁷) have been successfully stimulated by taking advantage of nanoparticles / nanofibers showing piezoelectric behavior. Indeed, these nanomaterials are able to generate electricity in response to mechanical deformations, which can be accomplished by using ultrasounds (US).¹⁸ In this context, here we suggest the use of the TPL technique for the fabrication of 3D piezoelectric structures (specifically the above mentioned Osteo-Prints doped with piezoelectric nanoparticles) in order to further promote and enhance the osteogenic differentiation of SaOS-2 cells.

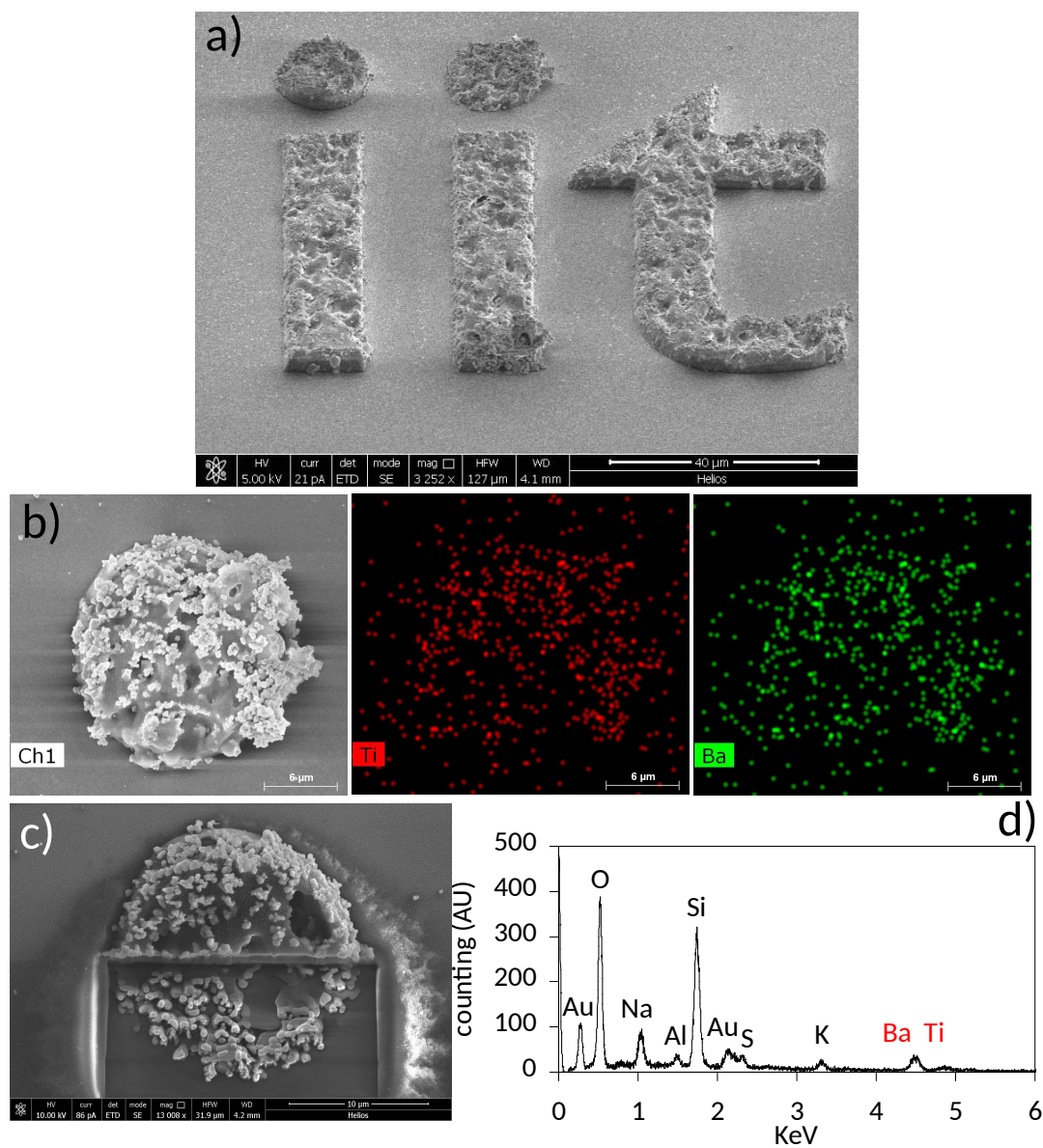


Figure 1. 3D nanocomposite (Ormocomp® resist / BTNPs 10 wt.%) structures fabricated by two-photon lithography (TPL). (a) The logo of our institute (IIT) was fabricated with a slice-by-slice approach and then imaged with scanning electron microscopy (SEM). (b) The left image is the SEM high-magnification scan of the dot of the letter "I" of the logo; center and left images show the energy-dispersive X-ray analysis (EDX) related, respectively, to the Ti (in red) and Ba (in green) elements, demonstrating the presence of the BTNPs on the surface of the structure. (c) A focused ion beam was used to mill the structure and analyze its inner composition: the SEM scan (c)

reveals the presence of the nanoparticles embedded in the polymerized structure; EDX spectrum (d) confirms the presence of the characteristic Ba and Ti peaks at 4.44 and 4.50 keV, respectively.

The preparation of the nanocomposite photoresist was carried out by mixing the commercially available biocompatible Ormocomp® resist with piezoelectric barium titanate nanoparticles (BTNPs, 10 wt.%). In order to test the obtained nanocomposite photoresist, TPL of a 3D structure resembling the logo of our institute (IIT) was fabricated through the slice-by-slice approach. Fig. 1a shows a scanning electron microscopy (SEM) tilted scan of the obtained logo, the surface of which appears particularly rough due to the presence of a large amount of nanoparticles. Single BTNPs on the surface of the IIT logo can be appreciated in Fig. 1b. In particular, the left image is a higher magnification SEM acquisition of a small portion of the previous structure. Energy-dispersive X-ray analysis (EDX) performed on the same scan clearly confirm the presence of nanoparticles on the structure surface, highlighted by the co-localized presence of Ti (in red, image in the middle) and of Ba (in green, image on the right). In order to assess the presence of the nanoparticles in the whole volume of the structure, a focused ion beam was used to mill the structure and analyze its inner composition. The SEM scan (Fig. 1c) revealed the presence of the nanoparticles embedded in the polymerized structure, confirmed by the EDX analysis, reported in Fig. 1d, that highlighted the presence of the characteristic Ba and Ti peaks at 4.44 and 4.50 keV, respectively (other peaks are related to the silica substrate, to the gold sputtering, and to other elements composing the resist). All together, SEM and EDX characterizations demonstrate the possibility to obtain, by TLP and with high reproducibility, 3D structures composed by Ormocomp® and piezoelectric BTNPs (both localized on the surface and embedded inside the structures).

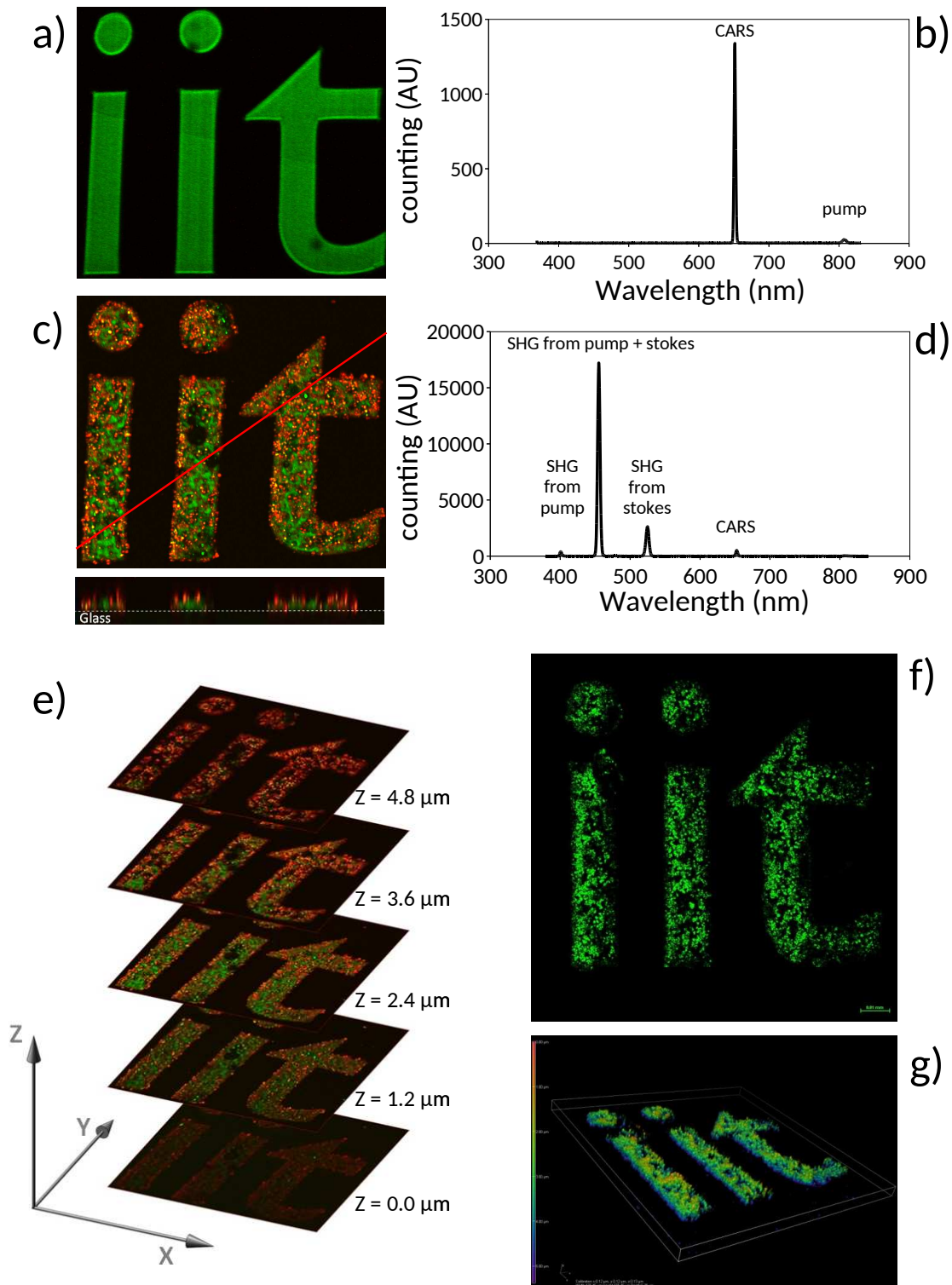


Figure 2. 3D optical characterization of the nanocomposite structures. In (a-e) the multi-modal (CARS and SFG) non-linear optical imaging is reported; the CARS signal (emitted by CH₂ bonds) is mapped in green and the SFG

signal (emitted by the BTNPs) in red. (a) *XY* optical section (140 μm x 140 μm) and (b) emission spectrum of the "control" logo (plain Ormocomp®). (c) *XY* optical section (140 μm x 140 μm) and (d) emission spectrum of the nanocomposite logo (Ormocomp® / BTNPs). (e) *XY* optical sections (140 μm x 140 μm) from a *z*-stack of the nanocomposite IIT logo. (f-g) Confocal laser scanning microscopy (CLSM) imaging of the nanocomposite logo. (f) Single *XY* optical section (BTNPs in green) and (g) 3D rendering of the nanocomposite logo (pseudo-colors have been used to code the depth along the *z* axis).

Nanoparticle distribution inside the structures has been investigated also with "non-destructive" methods, taking advantage of the peculiar nonlinear optical properties of the

¹⁹ BTNPs. These nanoparticles can be in fact easily imaged by multi-modal non-linear optical microscopy through the detection of the coherent anti-Stokes Raman scattering (CARS) and of the sum frequency generation signal (SFG, Figs. 2a-e), and by standard confocal laser scanning microscopy (CLSM, Fig. 2f-g). ²⁰ Using the multi-modal non-linear optical imaging setup, we visualized both structures polymerized with plain Ormocomp® (without BTNPs, Figs. 2a), and fabricated with the BTNP-doped resist (Figs. 2c, e). Exploiting a Stokes and a pump-and-probe beam (PaPB) exciting lasers at 1060 nm and 810 nm, respectively, we detected the CARS radiation emitted by the CH₂ bonds of the Ormocomp® at 655 nm (in green in the images, emission spectrum reported in Fig. 2b) and the SFG radiation emitted by the BTNPs at 460 nm

(in red in the images, emission spectrum reported in Fig. 2d). ^{21,22} The presence of the nanoparticles inside the structure was confirmed by a *z*-stack acquisition (Fig. 2e). Exploiting the CLSM setup, instead, we imaged the BTNPs inside the structure by using an excitation laser at 642 nm and by collecting emitted signal from 670 to 750 nm, as demonstrated by a single slice

of a *z*-stack acquisition (Fig. 2f) and by the relative 3D rendering (Fig. 2g, pseudo-colors have been used to code the depth along the *z* axis).

The investigation of the material piezoelectricity was performed through piezoresponse force microscopy (PFM). This technique consists in applying an alternating voltage to the cantilever tip while simultaneously measuring the material deformations as a result of the converse piezoelectric effect; further details regarding the adopted PFM system are reported in the Supporting Information. PFM analysis confirms that the nanocomposite material is characterized by a significantly higher d_{33} (0.57 ± 0.08 pm/V) compared to that one of the plain Ormocomp® without BTNPs (0.07 ± 0.01 pm/V): this result suggests that the presence of the BTNPs confers piezoelectric properties to the structure, confirming the success of the proposed approach.

In order to promote the osteogenic differentiation by combining topographic and piezoelectric stimulations, Osteo-Prints bioinspired by the 3D shape of the bone trabeculae were obtained through the slice-by-slice TPL of the described nanocomposite photoresist (Ormocomp® / BTNPs). In Fig. 3a, SEM imaging confirms the success of the polymerization process, having obtained structures of well-defined shapes doped with piezoelectric BTNPs, clearly observable at higher magnifications (from left to right increasing magnification SEM images are reported).

Square matrices of repeated Osteo-Prints (located at a 50 μm distance from each other) have been developed for extensive biological testing on human osteoblast-like SaOS-2 cells. Fig. 3b depicts CLSM acquisitions of SaOS-2 grown for 24 h on the Osteo-Prints: in green we can appreciate the presence of the BTNPs, in red the f-actin of the cell cytoskeleton (stained with TRITC-phalloidin) and in blue the nuclei (DAPI). These images qualitatively reveal as cells make strict connections to the surface of the scaffolds, by conforming their shape to that of the scaffolds (as indicated by white arrow in the images), as already reported in our previous work;⁴

furthermore, they are characterized by elongated/curved-shaped nuclei. As already pointed out, several investigations stressed that nucleus elongation, induced by cell deformation, is associated with chromatin condensation, and thus plays a key role during the process of cellular differentiation, representing a critical step in the mechanotransduction pathway that transduces the topographical cues into altered genic expression.

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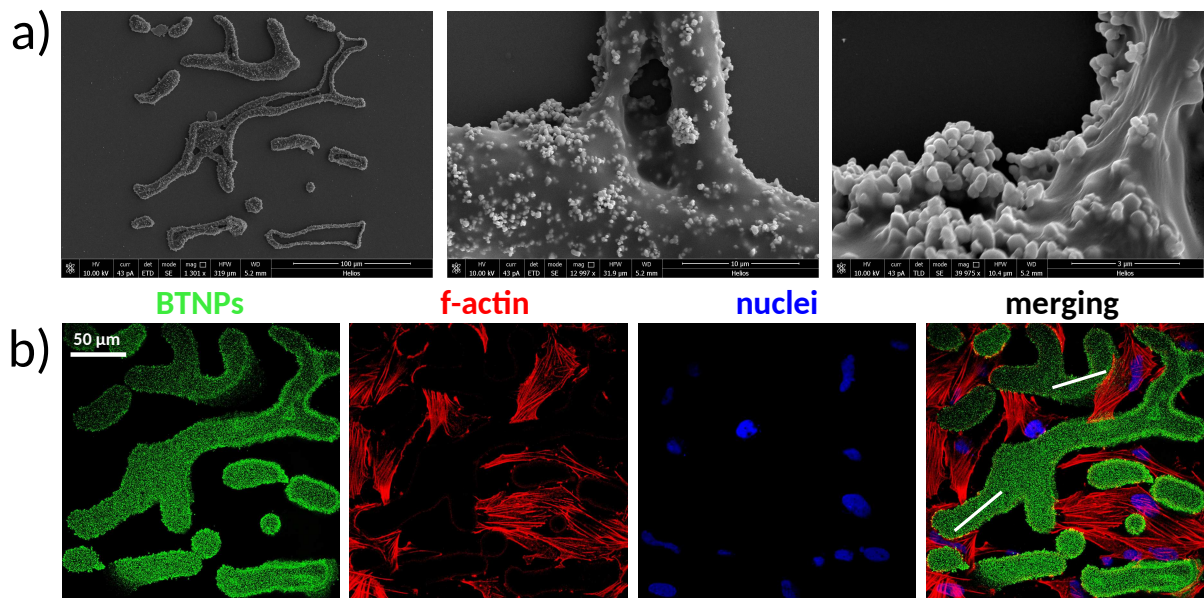


Figure 3. (a) SEM imaging of a nanocomposite Osteo-Print (left); high magnification images of the Osteo-Print (in the center and in the right) show the presence of the BTNPs on the surface of the 3D structure. (b) Confocal laser scanning microscopy of SaOS-2 osteoblast-like cells cultured on a matrix of repeated Osteo-Prints; BTNPs in green, f-actin in red, nuclei in blue. White arrows highlight strong interaction cells/scaffolds, affecting cellular shape.

After 24 h from the cell seeding, osteogenesis was induced for 3 days in SaOS-2 cells by using a culture medium supplemented with β -glycerophosphate, dexamethasone, and ascorbic acid

(please see the Supporting Information for the details about cell culture / differentiation). During the differentiation, cells were divided into four experimental groups: cells differentiated on *i*) control plain Osteo-Prints (OP) (non-doped), *ii*) Osteo-Prints doped with piezoelectric nanoparticles (OP/BTNPs), *iii*) plain Osteo-Prints stimulated with ultrasounds (OP + US), and *iv*) doped Osteo-Prints stimulated with ultrasounds (OP/BTNPs + US). The US stimulation was performed at 0.8 W/cm^2 for 5 s every 4 h, 3 times a day, during the all differentiation period.

After the stimulation protocol, immunocytochemistry experiments were performed against two markers (Fig. 4a, first and second row), Ki-67 and collagen type 1 (COL1). Ki-67 is a nuclear marker expressed in proliferating cells during G1, S, G2 and M cell cycle phases. Interestingly, the percentage of SaOS-2 cells expressing Ki-67 in response to the OP/BTNPs + US treatment ($40 \pm 3 \%$) was significantly lower compared to the other experimental groups (Fig. 4b, first plot): OP ($71 \pm 5 \%$), OP + US ($62 \pm 5 \%$) and OP/BTNPs ($64 \pm 4 \%$, Fig. 4b). This finding highlights a promotion of the exit from cell cycle when cells underwent to the synergic stimulation (OP/BTNPs + US). Moreover, synergic stimulation also affected the expression of

COL1 protein, a marker highly up-regulated during osteogenesis.²⁴ Percentage of area occupied by the collagen deposits produced by SaOS-2 differentiated in the OP/BTNPs + US condition was significantly higher ($0.7 \pm 0.2 \%$) compared to the other experimental groups (Fig. 4b, second plot), namely OP ($0.3 \pm 0.1 \%$), OP + US ($0.3 \pm 0.1 \%$) and OP/BTNPs ($0.2 \pm 0.1 \%$).

Finally, osteogenesis progression can be also reliably monitored as function of the hydroxyapatite (HA) nodules deposition.²⁵ Osteoimage™ assay was performed to investigate the bone mineralization efficiency (further experimental details available in Supporting Information), in terms of HA deposition in the SaOS-2 cultures (Fig. 4a, third row) . The

percentage area (average \pm standard deviation) occupied by the HA deposits in the 4 experimental groups is reported in the third plot of Fig. 4b. The HA deposit area of SaOS-2 in OP + US (8.5 ± 4.0 %) and OP/BTNPs (7.5 ± 2.5 %) cultures is higher, but not significantly different with respect to that one of the OP control experimental group (2.4 ± 1.7 %, $p > 0.05$). Instead, the HA deposition in the OP/BTNPs + US group (12.2 ± 3.3 %) was found significantly higher compared to the controls (OP, $p < 0.01$). All together, these results indicate that both the presence of the BTNPs in the structures and the US stimulation concur to induce a significant osteogenesis enhancement.

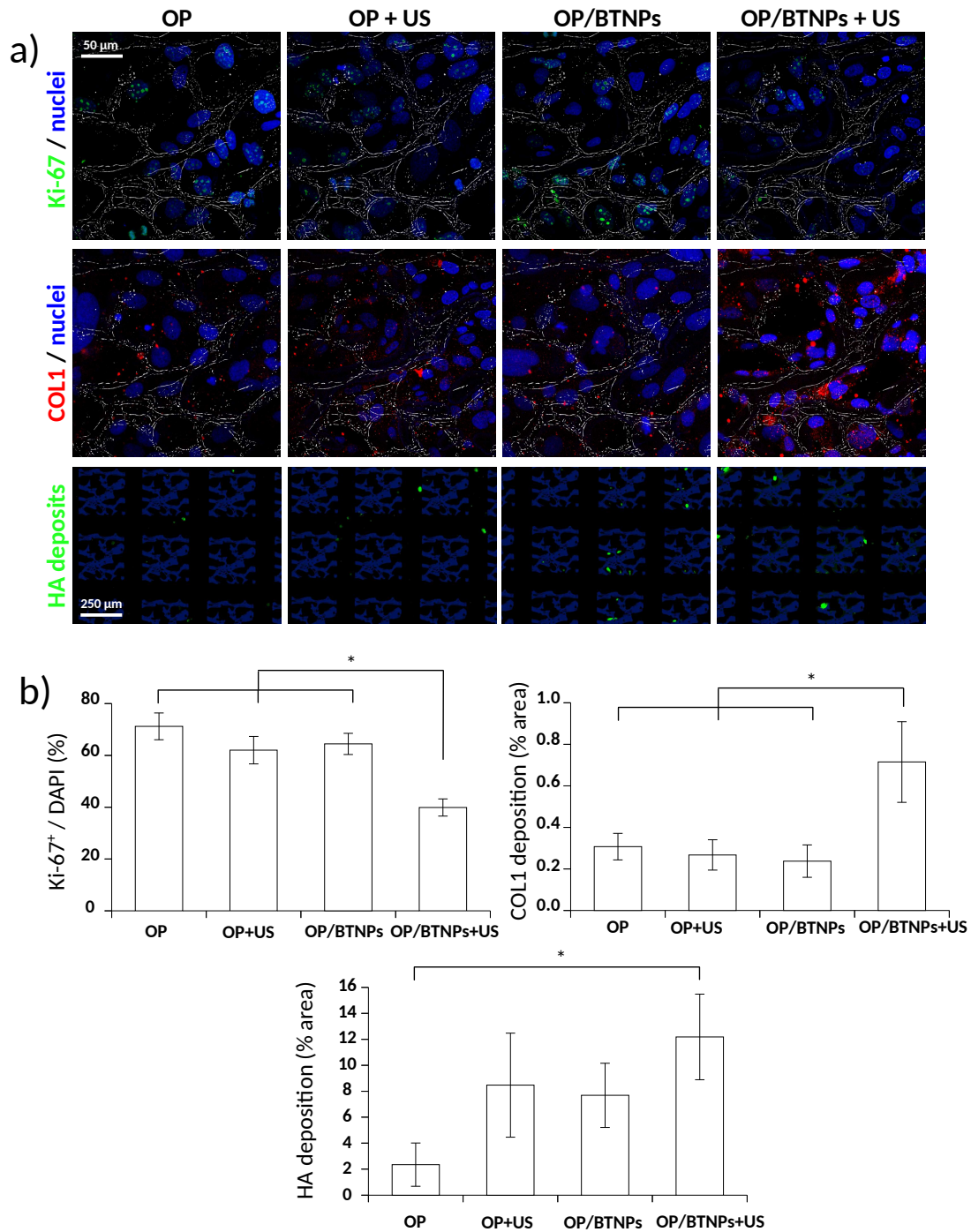


Figure 4. Nanocomposite bioinspired Osteo-Prints for the promotion of the osteogenic differentiation. (a) Confocal laser scanning microscopy images of Ki-67 and collagen type 1 (COL1) and fluorescence microscopy images of hydroxyapatite in SaOS-2 cultures differentiated on control Osteo-Prints (OP), on nanocomposite Osteo-Prints (OP/BTNPs), and that underwent ultrasound stimulation on the two structures (OP + US and OP/BTNPs + US). Ki-

67 in green, COL1 in red, nuclei in blue, and hydroxyapatite in green. (b) Quantitative analyses of the above described markers (average \pm standard deviation, * $p < 0.01$).

It is possible to assume that a piezoelectric excitation is involved in mediating these effects: indeed, in a recent work we demonstrated as BTNPs, activated with the same US intensities, are

able to piezoelectrically stimulate neural cells.²⁰ In this context, it is well known from the

literature that electric stimulation is able to promote osteogenesis, both *in vitro*²⁶ and *in vivo*,²⁷ thus a realistic scenario envisages an US-activation of the BTNPs on the OP, that are able to give electric stimuli to the cells, thus enhancing the differentiation process. However, besides piezoelectricity, other phenomena may contribute to the observed osteogenesis enhancement, as a temperature increase and the mechanostimulation of the US, or the roughness due to the presence of the BTNPs on the OP surface. Concerning mechanical stimulation, several examples in the literature suggest the important role played by this physical cue as an anabolic agent for

bone regeneration.²⁸ Among mechanical stimuli, US in particular result able of inhibiting the adipogenic differentiation and promoting osteogenesis of mesenchymal stem cells by acting on

the ROCK-Cot/Tpl2-MEK-ERK pathway.²⁹ Further efforts will be thus devoted in future works in order to deeply investigate the role of the different mechanisms mediating the osteogenic enhancement in response to the synergic OP/BTNPs + US stimulation.

Concluding, we exploited for the first time the TPL technique to fabricate bioinspired 3D structures (Osteo-Prints) with a nanocomposite resist (Ormocomp® / BTNPs). Complementary methods (SEM, FIB, EDX, CARS, SFG, CLSM and PFM) allowed a deep characterization of

the obtained structures. These scaffolds, in concomitance with a mechanical stimulation provided by US, resulted able to enhance the osteogenic differentiation of SaOS-2 bone-like cells.

Obtained findings open new interesting perspectives in the field of regenerative medicine and bone tissue engineering: as an example, the possibility to functionalize the surface of biomedical devices with biomimetic piezoelectric structures can potentially foster the osseointegration of the implants, by combining the US-driven piezoelectric stimulation with the presence of a bioinspired topography. This is a unique opportunity that is not offered, for example, by traditional titanium- and carbon-based materials: BTNP-based nanocomposites, with respect to

the other nanostructured materials adopted for bone tissue engineering,³⁰ not only mimic the piezoelectric properties of the bone, but they can be also remotely actuated for promoting the osseointegration / bone regeneration.

Supporting Information Available. More details of the experimental methods used are available as Supporting Information. Paragraphs concerning structures fabrication and characterization, biological experiments and relevant analysis are provided.

Acknowledgements

This research has been partially supported by the Italian Ministry of Health, Grant Number RF-2011-02350464.

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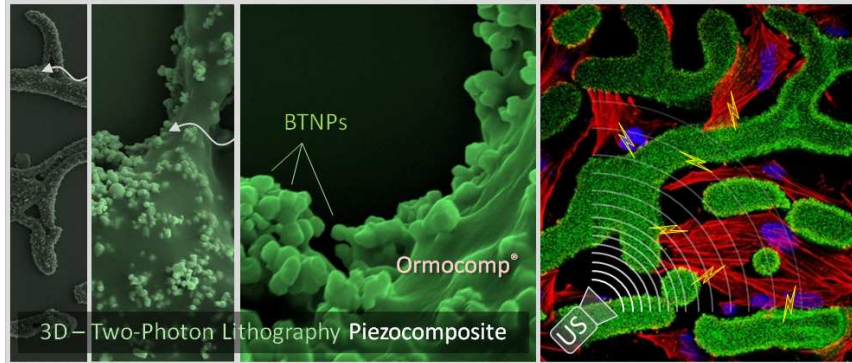
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SUPPORTING INFORMATION

Two-Photon Lithography of 3D Nanocomposite Piezoelectric Scaffolds for Cell Stimulation

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Experimental

Two photon lithography (TLP)

The two-photon lithography (TLP) system used in this work (Photonic Professional, Nanoscribe GmbH) is combined to an inverted optical microscope (Zeiss) with a 100× magnification oil-immersion objective, characterized by a numerical aperture $NA = 1.4$. Photons at wavelength of 790 nm and with a pulse duration of 120 fs were used with a laser power of 30 mW and an average scan speed of 50 $\mu\text{m/s}$ in order to cure, on a glass coverslip, the plain Ormocomp® and the Ormocomp® doped with 10 wt.% of barium titanate nanoparticles (BTNPs, Nanostructured and Amorphous Materials, for an extensive characterization please refer to our previous work¹). BTNPs were homogeneously mixed in Ormocomp by sonication at 8 W for 2 min with the aid of a tip sonicator (Ultrasonic Homogenizer 7 mini 20, Bandelin). The structures have been obtained through the slice-by-slice approach. Finally, the non-polymerized photoresist was removed by washing glass dishes with Ormodev® and isopropanol.

Structure characterization

Structures fabricated by TLP were characterized by using different techniques: scanning electron microscopy (SEM) equipped with energy-dispersive X-ray (EDX) spectroscopy, focused ion beam (FIB) milling / imaging, coherent anti-Stokes Raman spectroscopy (CARS), sum frequency generation (SFG) microscopy, confocal laser scanning microscopy (CLSM) and piezoresponse force microscopy (PFM).

Before SEM imaging, the samples were previously gold-sputtered for 25 s at 60 nA (RF/DC Magnetron Sputtering, Kenosystec). Subsequently, scans were acquired by using a Helios NanoLab 600i FIB/SEM, FEI system. FIB (FEI Helios 600) was performed to mill part of the structure with a milling current of 2 pA and an accelerating voltage of 30 kV.

CARS signal has been exploited to visualize the Ormocomp® CH₂ bonds, while BTNPs have been detected thanks to their strong sum frequency generation (SFG) signal. The 810 nm pump-and-probe beam (PaPB) is generated by a Ti-Sa femtosecond pulsed laser (Chameleon Vision 2, Coherent Inc.). Part of this beam is used to pump an optical parametric oscillator (OPO, Oria IR, Radiantis) that generates the 1060-nm signal beam used as the Stokes beam. Each beam passes through a telescopic beam expander and a SF6 optical-glass block, the thicknesses of which were carefully chosen in order to achieve spectral focusing in the sample optical plane by optimal tuning of the pulse group delay dispersion. The two beams are then overlapped in space (by means of a 900 nm long-pass dichroic mirror) and in time (using a delay-line). The combined PaPB and Stokes beam are then routed to the high-numerical-aperture lens (Objective C-Achroplan W, 32×, $NA = 0.85$, Carl Zeiss MicroImaging GmbH) of an inverted microscope (Axio Observer Z1, Carl Zeiss MicroImaging GmbH) through a pair of galvo-scanning mirrors (GVS002, Thorlabs), a scan lens, and a tube lens. The 655 nm CARS signal originated from the sample is collected in the trans-direction by a condenser lens ($NA = 0.55$), filtered (filter centered at 650 nm and with a full width at half maximum of 50 nm) and then detected by means of red-sensitive photomultiplier tube (R3896, Hamamatsu). The 460 nm SFG signal originated from the sample is collected in the epi-direction and routed toward the detector by means of a dichroic mirror. The signal is filtered (filter centered at 450 nm and with a full width at half maximum of 40 nm) and then focused to a second photomultiplier tube (R3896, Hamamatsu) by an additional

lens. For the spectral measures, the light collected by the condenser in trans-direction was edge-filtered to remove the pump photons and then routed to a spectrometer (iHR550, with Synapse sCCD camera, Horiba) by means of an optical fiber. The imaging data were processed with ImageJ software (U.S. National Institutes of Health) and the spectral data were processed with SynerJY software (Horiba).

Confocal laser scanning microscopy (CLSM, C2s, Nikon) was alternatively adopted for detecting BTNPs in the structures by using a laser excitation of 642 nm and by collecting the emission from 670 to 750 nm.

Direct piezoelectric coefficient d_{33} has been estimated by measuring the converse piezoelectric effect with PFM technique.² The bottom and top electrodes were respectively obtained by curing the samples (Ormocomp® and Ormocomp® / BTNPs) on gold-coated coverslips, and by the subsequent gold sputtering of the upper surface of the samples. An *a.c.* voltage excitation $V = V_0 \cdot \cos(\omega_{res}t)$ has been directly applied to the samples through gold electrodes. Sample surface oscillations have been detected with an atomic force microscope (AFM) in contact mode. The oscillation amplitudes of the AFM cantilever, A , has been demodulated at the resonant frequency ω_{res} of the cantilever that is still exhibited in contact conditions (contact resonance mode²), to improve measurement sensitivity. The demodulated oscillation amplitude, $A(\omega_{res})$, was used for the estimation of the direct piezoelectric coefficient by considering $d_{33} = A(\omega_{res}) / QV_0$, with Q the quality factor of the contact resonance. Measures have been carried out with a MultiMode AFM equipped with a Nanoscope IIIa controller (Veeco Instruments). Voltage excitation and signal demodulation have been performed using a SR830 DSP lock-in amplifier (SRS™) controlled by a home-made Labview software. Silicon AFM probes Nanoworld ARROW-CONT-50 have been used, with nominal free resonant frequency $f = 14$ kHz, and

nominal elastic constant $k = 0.2$ N/m. The contact resonance for this cantilever was about 80 kHz, and its quality factor about 50. The same equipment was used for the AFM surface imaging of the sample topography.

Cell culture and biological testing

SaOS-2 human osteosarcoma cells (ATCC HTB-85) were cultured in T25 flasks with Dulbecco's modified Eagle's medium (DMEM) supplemented with fetal bovine serum (10%), penicillin (100 U/ml), streptomycin (100 μ g/ml), and L-glutamine (2 mM). SaOS-2 cells were detached by using trypsin 0.05% with 0.02% EDTA and then plated on glass coverslips presenting the Osteo-Prints at a density of 10^4 cells/cm². Proliferation medium was replaced with differentiation medium after 24 h since seeding. The composition of the differentiation medium was low-glucose DMEM supplemented with fetal bovine serum (10%), dexamethasone (100 nM), β -glycerophosphate (10 mM), ascorbic acid (50 μ M), penicillin (100 U/ml), streptomycin (100 μ g/ml), and L-glutamine (2 mM). Osteogenesis was inducted for 3 days before biological testing. SaOS-2 adhesion on scaffolds was analyzed at 24 h since plating. Samples were fixed by using 4% para-formaldehyde in PBS for 20 min at 4°C. Subsequently, f-actin was stained with TRITC-conjugated phalloidin (Sigma, 1:100) and cell nuclei counterstained with DAPI (Millipore, 1:1000), according to standard staining procedures.

Differentiation samples were fixed in PFA as previously described, and immunocytochemistry against Ki-67 and collagen type 1 (COL1) and OsteoimageTM bone mineralization fluorescence assay (Lonza) were performed. Concerning immunostaining procedures, protocols were followed according to previous works.³ In particular, a primary rabbit anti-Ki-67 IgG (1:300 antibody dilution in 10% goat serum, Millipore), a primary rabbit anti-COL1 IgG (1:100 in 10% goat serum, Millipore), and a secondary goat anti-rabbit FITC-IgG (1:250 in 10% goat serum,

Molecular Probes) were used. Samples were observed and analyzed with a C2s confocal laser scanning microscope (Nikon). Osteoimage™ assay allowed the detection of the bone-like nodules deposited by the SaOS-2 cultures. Hydroxyapatite (HA) portion of the bone-like deposits was fluorescently labeled following the manufacturer's protocol, and subsequently detected by epifluorescence microscopy (Eclipse Ti-E fluorescence microscope, Nikon) with a low-magnification objective (10×).

Image measurements and statistical analysis

Biological triplicates were carried out, and at least 500 cells and 250 HA deposits were examined *per* experimental condition by using ImageJ. The percentage of the area occupied by deposits of HA or COL1 was considered for statistical comparisons. Normality of the distributions was verified with the Shapiro normality test and, subsequently, the ANOVA parametric test followed by the Tukey's HSD *post-hoc* test was performed through *R* software. Distributions were plotted on a bar graph showing media \pm standard deviation. Differences were considered significant at $p < 0.05$.

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